

# The Effects of Primary Structures on Photo-Induced Insolubilization of All-Cis Poly(*p*-phenylenevinylene)s in Thin Films

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All-cis poly(*p*-phenylenevinylene)s (PPVs) prepared by Suzuki–Miyaura-type polycondensation of (*Z,Z*)-1,4-bis(2-bromoethenyl)benzene with 2,5-dioctyloxybenzene-1,4-diboronic acid undergo photo-induced insolubilization in thin films, along with cis-to-trans isomerization of vinylene linkages to give all-trans PPVs. This phenomenon has been investigated using all-cis and all-trans PPVs with a range of molecular weights and terminal structures. The amounts of PPVs remaining on quartz substrates, after UV-irradiation on spin-coated films, followed by rinsing twice with CHCl<sub>3</sub>, are dependent on the geometries of vinylene linkages, molecular weights, and terminal structures of the starting PPVs. It has been found that the all-cis configuration is of particular importance for insolubilization. The insolubilized films exhibit charge-carrier mobility ( $\mu_{\text{FET}}$ ) up to  $1.6 \times 10^{-4} \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ .

$\pi$ -Conjugated polymers are promising materials for the development of flexible optoelectronic devices such as organic field-effect transistors (OFETs) and organic light-emitting diodes (OLEDs).<sup>1</sup> Poly(*p*-phenylenevinylene)s (PPVs) are among the most studied.<sup>2</sup> A distinct advantage of polymeric materials is the accessibility of thin films by solution processes such as spin-coating and dip-coating, which enable easy and low-cost fabrication of devices. Since it is difficult to gain high device performance with a single component, most devices have multilayer structures. In this connection, several methods of insolubilizing thin films have been developed to avoid interdiffusion of polymer layers upon multi-coating; the establishment of well-defined interfacial boundaries of thin films is an essential requirement for achieving high device performance.<sup>3</sup> The most extensively investigated method is the use of photo- or thermal-cross-linkable polymers.<sup>4–14</sup> Alternatively, soluble precursors coated onto substrates are converted to insoluble polymers by thermal reactions.<sup>2</sup> Although insoluble films of PPVs are generally prepared by the latter method, it is known that the resulting polymers often involve structural defects due to incomplete reactions.<sup>15</sup>

We recently found that PPVs gain a unique film-forming property when the vinylene linkages are highly regulated to cis geometry.<sup>16,17</sup> All-cis PPV ( $M_n = 4700$ ,  $M_w/M_n = 1.92$ ) prepared by Suzuki–Miyaura-type polycondensation of (*Z,Z*)-1,4-bis(2-bromoethenyl)benzene with 2,5-dioctyloxybenzene-1,4-diboronic acid is highly soluble in organic solvents and forms an amorphous film by spin-coating.<sup>16c</sup> Interestingly, the resulting film becomes much less soluble under UV-irradiation, along with cis-to-trans isomerization of vinylene linkages to give all-trans PPV.<sup>16a,18</sup> Although one may assume that the photoisomerization giving all-trans PPV with a rigid framework causes this “photo-induced insolubilization” phenome-

non, it has been observed that MEH-PPV as a commercial product with the all-trans configuration<sup>21</sup> remains soluble under UV light.

To clarify the factors governing this interesting phenomenon, we prepared in this study all-cis and all-trans PPVs with a range of molecular weights and end-group structures by Suzuki–Miyaura-type polycondensation, and examined their performance in photo-induced insolubilization by a quantitative method. As a result, the all-cis configuration has proven to be a particularly important factor. It has also been found that a 2-bromoethenyl group at the chain-end facilitates the photo-induced insolubilization to a notable extent.

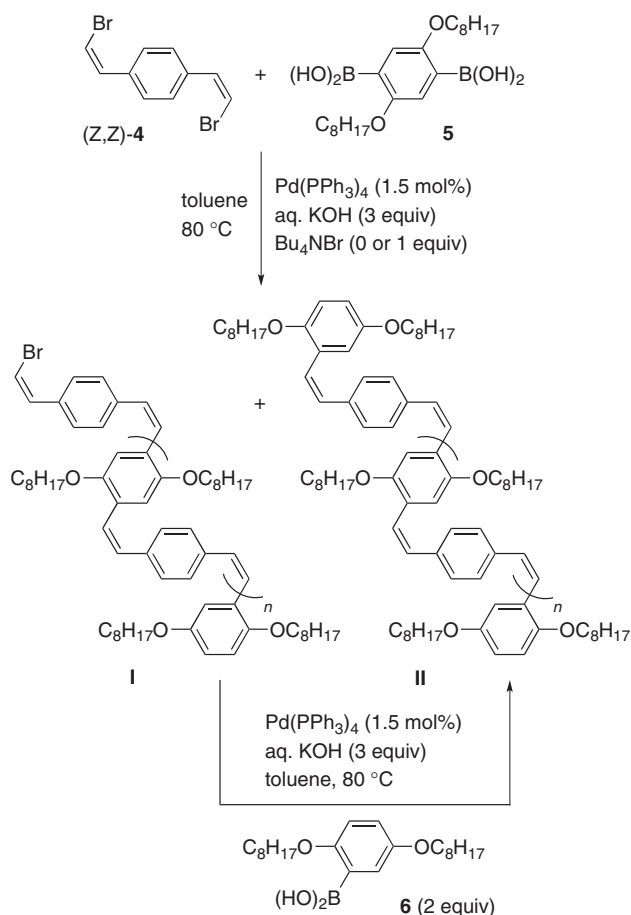
## Results and Discussion

**Preparation of All-Cis and All-Trans PPVs.** Table 1 lists the all-cis PPVs (**1a–1h** and **2a–2c**) and all-trans PPVs (**3a–3d**) employed in this study. Scheme 1 shows the synthetic routes to all-cis PPVs. We have already documented that polycondensation of (*Z,Z*)-1,4-bis(2-bromoethenyl)benzene (**4**) with 2,5-dioctyloxybenzene-1,4-diboronic acid (**5**) catalyzed by Pd(PPh<sub>3</sub>)<sub>4</sub> forms two types of all-cis PPVs having different terminal structures (i.e., I and II in Scheme 1).<sup>16b</sup> Type I polymer bears a 4-(2-bromoethenyl)phenyl and 2,5-dioctyloxyphenyl group at each terminus, whereas type II polymer has a 2,5-dioctyloxyphenyl group at each end. The catalytic reaction affords a product mainly consisting of type I polymer. Compounds **1a–1h** are classified into this sort. The molecular weights may be controlled by catalytic conditions, especially by the amount of Bu<sub>4</sub>NBr added to the system. The polymers with  $M_n$  of less than 3000 were synthesized in the absence of Bu<sub>4</sub>NBr, whereas those with higher molecular weights were obtained in the presence of Bu<sub>4</sub>NBr (1 equiv).

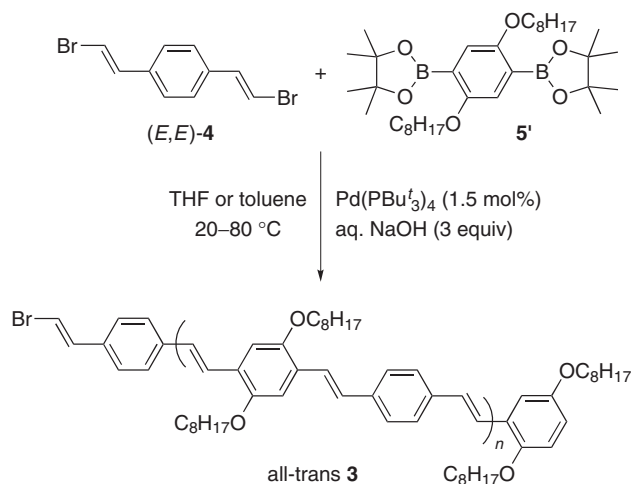
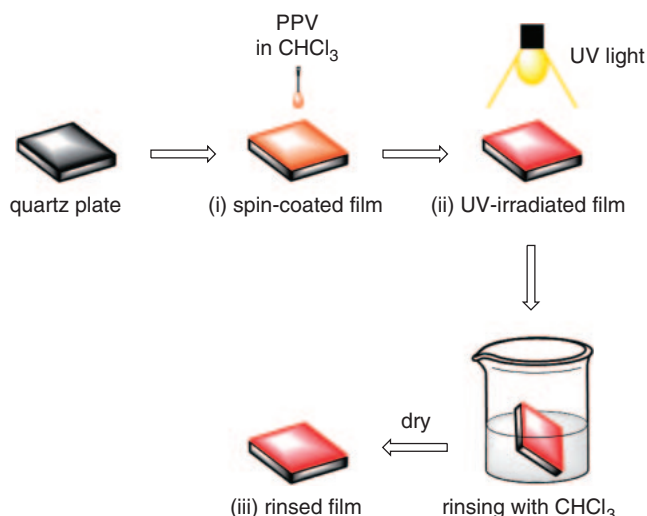
**Table 1.** List of PPVs and Their Performance in Photo-Induced Insolubilization

PPV (trans/cis)	$M_n^a$	$M_w/M_n^a$	Insolubilized PPV/%
<b>1a</b> (<1/99)	1800	1.39	36
<b>1b</b> (<1/99)	2800	1.55	33
<b>1c</b> (<1/99)	3500	1.55	59
<b>1d</b> (<1/99)	4700	2.79	56
<b>1e</b> (<1/99)	4800	2.20	64
<b>1f</b> (<1/99)	6600	3.14	85
<b>1g</b> (<1/99)	7000	2.69	84
<b>1h</b> (<1/99)	8100	3.93	89
<b>2a</b> (<1/99)	2400	1.24	2
<b>2b</b> (<1/99) <sup>b</sup>	5000	1.03	11
<b>2c</b> (<1/99) <sup>b</sup>	7700	1.11	35
<b>3a</b> (>99/1)	2200 <sup>c</sup>	1.57	5
<b>3b</b> (>99/1)	3300 <sup>c</sup>	1.44	10
<b>3c</b> (>99/1)	6100 <sup>c</sup>	1.80	22
<b>3d</b> (>99/1)	7900 <sup>c</sup>	1.96	43

a) Determined by GPC calibration based on polystyrene standards. b) Separated from **2a** by preparative GPC. c) Calibrated values to all-cis PPVs (see text).

**Scheme 1.** Synthetic routes to all-cis PPVs (**1** and **2**).

Type I polymer is converted to type II polymer by end-capping with 2,5-dioctyloxybenzeneboronic acid (**6**).<sup>16b</sup> This reaction was successful for the synthesis of **2a** with a relatively low molecular weight, but did not proceed well with higher

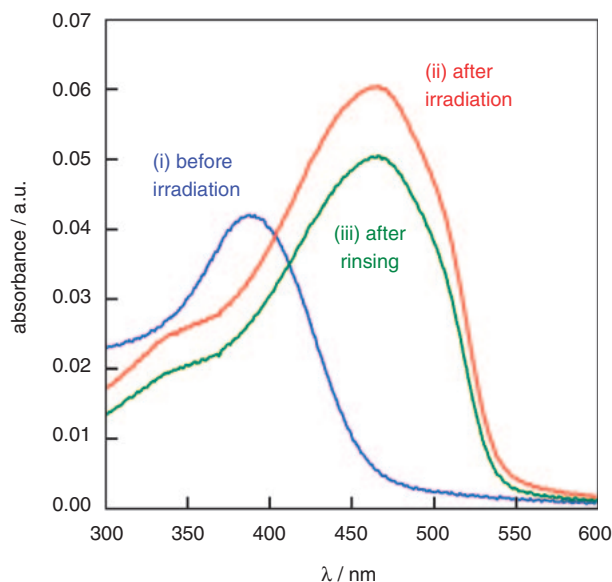
**Scheme 2.** Synthetic routes to all-trans PPVs (**3**).**Figure 1.** Experimental procedure for photo-induced insolubilization of PPVs.

molecular weight polymers. Thus, **2b** and **2c** in Table 1 were separated from **2a** by preparative GPC.

All-trans PPVs (**3a–3d**) were synthesized by Suzuki–Miyaura-type polycondensation of (E,E)-4 with the pinacol ester of **5** (**5'**) in the presence of Pd(PBu<sub>3</sub>)<sub>2</sub> as the catalyst (Scheme 2).<sup>20</sup> The resulting polymers bear the same end-group functionalities as type I polymers as confirmed by NMR spectroscopy. The  $M_n$  values given in Table 1 are calibrated to all-cis PPVs by using reference samples derived from all-cis PPVs by photoisomerization in solution, to avoid overestimation in GPC analysis arising from the rigid framework of all-trans PPVs.<sup>22</sup>

**Photo-Induced Insolubilization of PPVs.** Figure 1 illustrates the experimental procedure for photo-induced insolubilization. (i) A thin film of all-cis or all-trans PPV was prepared from a CHCl<sub>3</sub> solution (0.2 wt %) by spin-coating, and (ii) exposed to UV light ( $\lambda = 300–400$  nm, 21.0 mW cm<sup>−2</sup>) for 1 h at room temperature under vacuum. (iii) The irradiated film was rinsed twice with CHCl<sub>3</sub> and dried.

Figure 2 compares the UV–vis absorption spectra of the films at steps (i)–(iii) for all-cis PPV (**1g**). (i) The thin film of



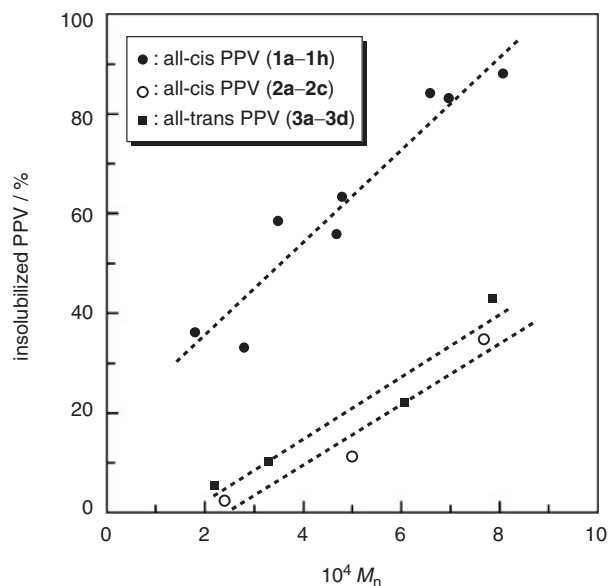
**Figure 2.** UV-vis absorption spectra of all-cis PPV (**1g**) in thin film, before and after UV-irradiation (i and ii), and after rinsing with  $\text{CHCl}_3$  (iii).

**1g** exhibited a  $\pi$ - $\pi^*$  transition at 388 nm. (ii) This absorption disappeared upon UV-irradiation, and a strong absorption assignable to the  $\pi$ - $\pi^*$  transition of all-trans PPV appeared at 466 nm instead. (iii) The absorption remained after rinsing, though the intensity decreased to a certain extent. In the case of **1g**, 84% of the absorption intensity of spectrum (ii) was preserved in spectrum (iii). We employed this intensity ratio as the performance index of PPV in photo-induced insolubilization (i.e., insolubilized PPV (%) in Table 1).

Similar experiments were performed with all-cis PPVs (**1a–1f** and **1h**), end-capped all-cis PPVs (**2a–2c**), and all-trans PPVs (**3a–3d**). It was confirmed by UV-vis absorption spectroscopy that all-cis PPVs were entirely converted to all-trans isomers irrespective of the molecular weights and terminal structures during the UV-irradiation for 1 h at room temperature.<sup>18</sup> On the other hand, all-trans PPVs remained in the all-trans form under UV light.<sup>20</sup>

Figure 3 plots the insolubilized PPV percentages as a function of number-average molecular weights of the starting PPVs. All-cis PPVs (**1a–1h**) possess much higher performance than all-trans PPVs (**3a–3d**). It is also obvious that the higher molecular weight polymer tends to remain in the higher percentage. Moreover, end-capped polymers (**2a–2c**) are clearly less efficient than **1a–1h** having a 4-(2-bromoethenyl)-phenyl group at the chain-end.

There is a possibility that the insolubilization in thin films is caused by cross-linking of polymer chains via photochemical reactions such as [2 + 2] cycloaddition of vinylene units or radical addition of C–Br bonds to vinylene linkages. In any case, the shortening of effective  $\pi$ -conjugation length of PPV chain should occur to a considerable extent, but no notable blue shift was observed for insolubilized films. Accordingly, we presumed supramolecular interaction as an alternative driving force for insolubilization. We have already confirmed that all-cis PPV cast from a solution remains amorphous for a few hours.<sup>16c</sup> The amorphous all-cis PPV has a low  $T_g$  (ca. 0 °C)



**Figure 3.** The relation between insolubilized PPV percentages and number-average molecular weights of PPVs.

and is in the liquid state at room temperature. Thus, all-cis PPV molecules are expected to be highly mobile in thin films. In this situation, all-trans PPV molecules generated from all-cis PPVs by photoisomerization possibly form a highly aligned structure with the aid of  $\pi$ - $\pi$  stacking between polymer chains, giving rise to the insolubilization phenomenon. On the other hand, it has been confirmed that PPVs prepared in the all-trans form from the beginning (i.e., **3a–3d**) have a higher  $T_g$  (ca. 50 °C) and are much less mobile in the solid state at room temperature,<sup>16c</sup> and this would be a prime reason for the much lower performance of **3a–3d**.

Although the above discussion is seemingly inconsistent with the low performance of the end-capped polymers (**2a–2c**), since halogen atoms on aromatic molecules are known to facilitate the formation of  $\pi$ -stacks,<sup>23</sup> it is possible that the photo-induced insolubilization is assisted by 4-(2-bromoethenyl)phenyl groups at the chain-end.

**OFET Properties of Insolubilized PPVs.** To gain experimental support for the occurrence of molecular alignment during the photo-induced insolubilization, charge-carrier mobility of PPV films was measured by using OFET devices fabricated in a top-contact configuration on Si/SiO<sub>2</sub> substrates. Molecular alignment of  $\pi$ -conjugated polymers has been known to be essential for the development of charge-carrier mobility.<sup>24,25</sup>

Table 2 summarizes the results. While all-cis PPVs simply coated on the substrates were insulators, they changed to semiconductors after UV-irradiation at room temperature for 1 h. The hole mobility was improved with increasing molecular weight of the starting polymer (Entries 1–4), and further by rinsing the film with toluene (Entry 4).<sup>26</sup> Interestingly, all-trans PPVs were totally insulators even after photoirradiation (Entries 5 and 6). Thus, these observations may be taken as another indication of the occurrence of molecular alignment during the photo-induced insolubilization of all-cis PPVs in thin films.

**Table 2.** Charge-Carrier Mobility of PPV Films before and after Photoirradiation

Entry	All-cis PPV		Mobility ( $\mu_{\text{FET}}/\text{cm}^2 \text{V}^{-1} \text{s}^{-1}$ )		
	$M_n^{\text{a}}$	$M_w/M_n^{\text{a}}$	Spin-coated film	Irradiated film	Rinsed film
1	5900	3.61	$<10^{-6}$	$1.0 \times 10^{-6}$	— <sup>c)</sup>
2	7700	2.69	$<10^{-6}$	$2.3 \times 10^{-5}$	— <sup>c)</sup>
3	11200	2.66	$<10^{-6}$	$5.0 \times 10^{-5}$	— <sup>c)</sup>
4	34600 <sup>b)</sup>	1.47	$<10^{-6}$	$5.0 \times 10^{-5}$	$1.6 \times 10^{-4}$
5	6100	1.80	$<10^{-6}$	$<10^{-6}$	— <sup>c)</sup>
6	7900	1.96	$<10^{-6}$	$<10^{-6}$	$<10^{-6}$

a) Determined by GPC calibration based on polystyrene standards. b) Separated from all-cis PPV with  $M_n = 15700$  by recycle GPC. c) Not determined.

### Conclusion

We confirmed that the all-cis configuration is of particular importance for photo-induced insolubilization of PPVs in thin films. The performance of all-cis PPVs is also dependent upon molecular weights and terminal structures. Thus, all-cis PPVs with  $M_n > 6600$  are insolubilized in 84–89% probability, when the polymers are equipped with a 4-(2-bromoethenyl)phenyl group at the chain-end. Although the insolubilization mechanism is not clear at present, the experimental observations reported in this paper include crucial information for designing poly(arylenevinylene)s that are insolubilized simply by UV-irradiation.

### Experimental

**General Considerations.** All manipulations using organo-metallic compounds were carried out under a nitrogen or argon atmosphere using conventional Schlenk techniques. Nitrogen and argon gas were dried by passing through  $\text{P}_2\text{O}_5$  (Merck, SICAPENT). Analytical GPC was carried out on a JASCO GPC assembly consisting of a model PU-980 precision pump, a model RI-1530 refractive index detector, and three polystyrene gel columns (Shodex KF-801, KF-803L, KF-805L). THF was used as the mobile phase with a flow rate of  $1.0 \text{ mL min}^{-1}$  at  $40^\circ\text{C}$ . The columns were calibrated against 9 standard polystyrene samples (Shodex;  $M_n = 980$ – $1920000$ ). Recycling preparative GPC was carried out on a JAI LC918U equipped with two preparative GPC columns (JAIGEL-1H-A and -2H-A).  $\text{CHCl}_3$  was used as the mobile phase with a flow rate of  $3.8 \text{ mL min}^{-1}$ .  $^1\text{H NMR}$  spectra were recorded in  $\text{CDCl}_3$  at  $25^\circ\text{C}$  on a Varian Mercury 300 spectrometer, operating at  $300.10 \text{ MHz}$ . Elemental analysis was performed by the ICR Analytical Laboratory, Kyoto University. Spin-coating of PPV was performed with a Mikasa spin coater 1H-DX2. Photoirradiation was carried out at room temperature with an Asahi Spectra LAX-101 Xe lamp. UV–vis absorption spectra were recorded on a JASCO V-560 spectrometer.

Toluene was dried over sodium benzophenone ketyl, distilled, and stored over activated molecular sieves (MS4A). The following compounds were synthesized according to the literature:  $\text{Pd}(\text{PPh}_3)_4$ ,<sup>27</sup> (Z,Z)-1,4-(2-bromoethenyl)benzene (**4**),<sup>28</sup> 2,5-diethyl-oxybenzene-1,4-diboronic acid (**5**),<sup>29</sup> 2,5-diethyloxybenzene-boronic acid (**6**),<sup>30</sup> and 1,4-bis(4,4,5,5-tetramethyl-1,3,2-dioxaborolan-2-yl)-2,5-diethyloxybenzene.<sup>16b</sup> All-cis PPVs (**1a–1h** and **2a**)<sup>16a,16b</sup> and all-trans PPVs (**3a–3d**)<sup>20</sup> were prepared as previously

reported. All other chemicals were obtained from commercial suppliers and used without further purification.

**Photo-Induced Insolubilization of PPVs.** A solution of PPV in  $\text{CHCl}_3$  (0.2 wt %) was passed through a syringe filter (DISMIC-13 JP, PTFE  $0.50 \mu\text{m}$ , Hydrophobic; ADVANTEC). A thin film of PPV was prepared by spin-coating on a quartz plate ( $1 \text{ cm}^2$ ); the filtrate ( $50 \mu\text{L}$ ) was placed dropwise on a plate, and the plate was accelerated to 1200 rpm for 2 s, kept at this rate for 10 s, and then at 2000 rpm for 60 s. After drying under vacuum at room temperature for 30 min, the film was placed in a quartz cell under a nitrogen atmosphere, and analyzed by UV–vis absorption spectroscopy. Next, the film was placed in a stainless-steel holder with a quartz window, irradiated by a Xe lamp ( $\lambda_{\text{max}} = 365 \text{ nm}$ ,  $21.0 \text{ mW cm}^{-2}$ ) for 60 min under vacuum at room temperature, and analyzed. Finally, the film was rinsed twice in  $\text{CHCl}_3$  (each 3 mL) with light shaking, dried under vacuum, and examined again by UV–vis absorption spectroscopy.

**Preparation of All-Cis PPVs with High Molecular Weights (Entries 3 and 4 in Table 2).** (Z,Z)-**1** (173 mg, 0.600 mmol), 1,4-bis(4,4,5,5-tetramethyl-1,3,2-dioxaborolan-2-yl)-2,5-diethyloxybenzene (352 mg, 0.600 mmol) and methyltriethylammonium chloride (Aliquat 336; 24.3 mg, 0.0600 mmol) were dissolved in toluene (3 mL), and aqueous  $\text{Na}_2\text{CO}_3$  (2.0 M, 0.90 mL, 1.8 mmol) and  $\text{Pd}(\text{PPh}_3)_4$  (10.4 mg, 9.00  $\mu\text{mol}$ ) were added. The mixture was vigorously stirred under reflux for 24 h in the dark. After cooling to room temperature, the mixture was diluted with  $\text{CH}_2\text{Cl}_2$  (8 mL), washed with water, and then poured into vigorously stirred  $\text{MeOH}/\text{CH}_2\text{Cl}_2$  (5/1, 180 mL). A yellow precipitate of all-cis PPV was collected by a membrane filter ( $0.5 \mu\text{m}$ ), washed with  $\text{MeOH}$ , and dried under vacuum at room temperature overnight (254 mg, 92% yield,  $M_n = 11200$ ,  $M_w/M_n = 2.66$ ). The polymer with  $M_n = 34600$  ( $M_w/M_n = 1.47$ ) was obtained from this product by preparative GPC.

**Measurement of Charge-Carrier Mobility.** All manipulations were carried out in an argon-filled glove box. An OFET device was fabricated in a top-contact configuration on a p-type Si wafer as a gate electrode (one-side polished,  $<0.02 \Omega$  resistance, Furuuchi Chemical Co.) with 3000-Å-thick  $\text{SiO}_2$  ( $C_i = 11 \text{ nF cm}^{-2}$ ) as a dielectric film. A plate was cut and washed in 2-propanol under ultrasonic wave and under boiling. A thin film of PPV (ca. 30-nm-thick) was deposited on the Si/ $\text{SiO}_2$  substrate by spin-coating from a 0.1–0.5 wt % solution in  $\text{CHCl}_3$  at 100–8000 rpm for 60 s, dried under vacuum at room temperature, and irradiated with a Xe lamp ( $\lambda_{\text{max}} = 365 \text{ nm}$ ,  $23.0 \text{ mW cm}^{-2}$ ) for 1 h. On the top of the thin film, gold films (30 nm) as source and drain electrodes were deposited through a mask. Drain-source channel length ( $L$ ) and width ( $W$ ) were  $20 \mu\text{m}$  and  $5 \text{ mm}$ , respectively. Characteristics of OFET devices were evaluated at room temperature using two source meters (2400, Keithley Instruments Inc.). Charge-carrier mobility ( $\mu_{\text{FET}}$ ) was calculated in the saturation regime ( $V_d = 100 \text{ V}$ ) of the  $I_d$  using the following equation,

$$I_d = (WC_i/2L)\mu_{\text{FET}}(V_g - V_t)_2 \quad (1)$$

where  $C_i$  is the capacitance of the  $\text{SiO}_2$  insulator, and  $V_g$  and  $V_t$  are the gate and threshold voltages, respectively. In another case, the thin film of PPV was rinsed twice with toluene (1 mL) after UV-irradiation, and examined in a similar manner.

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